
Room Temperature Magnetoelectric Coupling and AC Electrical Response of $0.2 \text{ NiFe}_2\text{O}_4 > 0.8 \text{ ZnO}$ Multiferroic Nanocomposite

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ABSTRACT

Multiferroic $0.2\text{NiFe}_2\text{O}_4 - 0.8 \text{ ZnO}$ nanocomposites have been prepared through low temperature chemical “pyrophoric reaction process” followed by solid state route. Structural characterization has been carried out through X-ray diffraction technique, which shows the co-existence of both the phases of the nanocomposites. The ac electrical properties of nanocomposites have been studied employing impedance spectroscopy technique. The impedance value is found to increase with increase in magnetic field attributing the magnetostriction property of the composites. Dielectric constant is found to decrease with increase in magnetic field. The magnetoelectric voltage has been obtained for both longitudinal and transverse direction. The value of magnetoimpedance and magnetodielectric is found to ~43 and ~ 34 % for magnetostriction property of the magnetostrictive materials.

Keywords

Nanocomposites, Impedance Spectroscopy, Dielectric, Magnetoelectric Coupling.

INTRODUCTION

Magnetoelectric multiferroics presenting coupled magnetic and ferroelectric order parameters have attracted a lot of attention towards the design of new materials to outfit the desired application. In semiconductor nanocrystal, there is an increasing research interest considering the issue of size miniaturization and surface state on nanometer scale [1]. Zinc Oxide (ZnO), hexagonal wurtzite structure, which is well known for its wide band gap of 3.37 eV and large excitonic binding energy of 60 meV, has been of growing technological interest due to its versatile favourable properties, achieved great variety of applications such as in data storage devices, photodetector, magneto-electric based devices, sensors or photocatalysis [2,3]. In particular, ZnO holding an interesting and preferential candidate for functioning as a multipurpose material that combines superior optical, chemical, electrical, piezoelectric and magnetic properties. Nowadays, multiferroic material constituting ferromagnetic and ferroelectric properties is a subject of intense research interest. Hence, it is essential to fabricate multiferroic composites for commercial applications in order to achieve multiferroicity and strong ME coupling. In this study, we have reported the structural characterization, magnetoelectric coupling and magnetic fields dependent dielectric and ac electrical properties of $0.2 \text{ NiFe}_2\text{O}_4$ (NFO) – 0.8 ZnO nanocomposites prepared by “chemical pyrophoric reaction process” followed by solid state route.

EXPERIMENTAL DETAILS

Multiferroic nanocomposites $0.2 \text{ NFO} - 0.8 \text{ ZnO}$ have been prepared by “chemical pyrophoric reaction process” [4,5] and solid state route. The NFO phase were prepared from high purity $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (Merck) and $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ (Merck). The ZnO phase were prepared from high purity $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$. After

preparation of NFO and ZnO nanoparticles, we have mixed both the nanoparticles in a requisite amount and grinded it for 10 h. Then the mixed powder has been calcined in air for 5 h at a temperature of 800°C to obtain the required nanometric composite phase. We have made circular pellets from calcined nano powders. Then these pellets have been used for electrical measurements.

Structural characterization has been done through X-ray diffraction technique (XRD) (Bruker D8 Advance Diffractometer) with monochromatic Cu-K_α radiation. Dielectric and ac electrical properties have been carried out using impedance analyzer (KEYSIGHT E4990A). For applied magnetic field, we have used electromagnet (Polytronic Laboratory Electromagnet, EM-100) and electromagnet power supply (Polytronic Power supply, CCP-100Y). We have made an electrode on the sample surface using high temperature silver paste. For ME measurement, we have used homemade ME measurement setup[6].

STRUCTURAL CHARACTERIZATION

Structural characterization has been carried out using x-ray diffractometer (XRD) with monochromatic Cu-K_α radiation at room temperature. Figure 1 shows the XRD pattern of 0.2 NFO– 0.8 ZnO nanocomposites, where the peaks are indexed using JCPDS software having JCPDS PDF No. 441485 and 030891 for NFO and ZnO nanocomposites, respectively. From XRD pattern, we have estimated the average crystallite size (Φ) for both NFO and ZnO phases of the composites using Scherrer formula as $\Phi = k\lambda/\beta\cos\theta$, where, λ is the wavelength of CuK_α radiation (1.542 Å), k is the shape factor, θ is the diffraction angle and β is the full width half maxima of the XRD peak. It is found that the average crystallite size for NFO phase to be ~ 52 nm ($\pm 1\%$) and for ZnO phase ~ 49 nm ($\pm 1\%$).

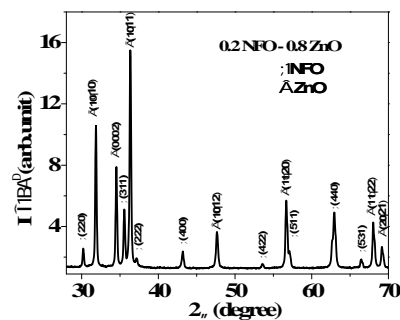


FIG 1: XRD pattern for 0.2 NFO- 0.8 ZnO nanocomposites.

ELECTRICAL PROPERTIES

The magnetic field dependent ac electrical properties of the composite have been investigated using impedance spectroscopy technique. Figure 2(a) shows the variation of Z as a function of frequencies at different magnetic fields. The value of the impedance is found to be in the order of kΩ range. The impedance (Z) has been studied at room temperature in presence of applied field (H) of 0, 800 and 2500 Oe. The real part (Z') and imaginary part (Z'') of impedance are measured on application of H in perpendicular direction of measured electric field. Z vs. f curves are found to gradually increase with increase in f at different applied H . The observed increased impedance value with increase in H may be due to the magnetostriction property of magnetostrictive materials [7,8]. We have estimated MI for this nanocomposites as shown in inset of Fig. 2(a). The magnetoimpedance (MI) is the change of impedance of a magnetic material under the application of H , which is related as [9] $MI (\%) = \frac{Z_H - Z_0}{Z_0} \times 100$ where, Z_H and Z_0 are the impedances under the application of finite H and absence of H , respectively. The observed MI is found to be positive, which is ~ 43% at f of 800 Hz. Furthermore, we have plotted magnetic field dependent Nyquist plots as shown in Figs. 2(b). Nyquist plots show two semi-circular arcs, one of which lies at high f region and the other lies at low f region. The observed peak at low f region corresponds to the grain boundaries of the sample, whereas the peak at high f region corresponds to the grain effect of the sample.

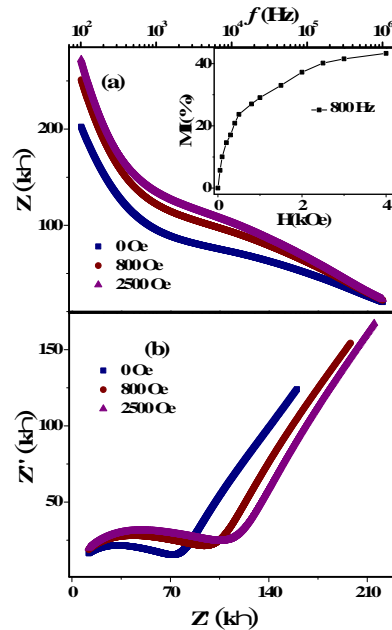


FIG 2: (a) Z vs.f,(b) Z' vs. Z at different magnetic fields, insets in (a) shows MI (%) vs.H plot.

Dielectric Properties

Dielectric constant (ϵ) as a function of f at different H has been measured as shown in Fig. 3(a). ϵ as a function of f is found to decrease with increase in H . At high f region, ϵ curves at different H are merged and behave as independent of f . High value of ϵ at low f region for different H may be explained through the light of Maxwell Wagner interfacial polarization[10]. Dielectric loss [$\tan \delta$] has been studied as a function of f at different H as shown in Fig. 3(b). Similarly, $\tan \delta$ also decreases with increase in H . Magnetodielectric (MD) response has been estimated using the relation as $MD(\%) = \frac{\epsilon(H,f) - \epsilon(0,f)}{\epsilon(0,f)} \times 100$, where, $\epsilon(H)$ is the dielectric constant in presence of H and $\epsilon(0)$ is the dielectric constant in absence of H . However, MD represents a strong coupling between piezoelectric and magnetostrictive phases of the nanocomposites[11]. $|MD(\%)|$ at a f of 500 Hz is found to $\sim 34\%$.

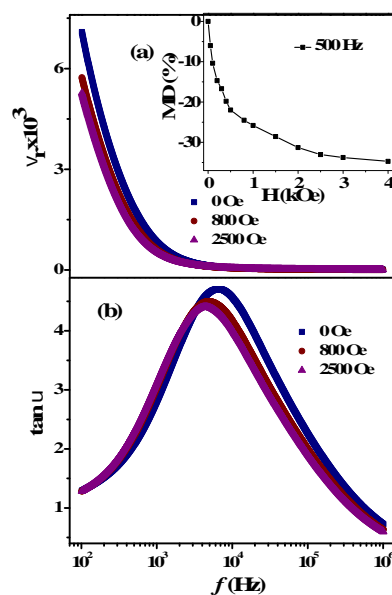


FIG 3: (a) ϵ vs.f,(b) $\tan \delta$ vs. f at different magnetic fields, inset in (a) shows MD (%) vs. H curves at $f = 500$ Hz.

MAGNETOELECTRIC EFFECT

Room temperature magnetoelectric effect (ϵ) as a function of H at af of 73 Hz in transverse and longitudinal configurations are shown in Fig. 4(a) and (b), respectively. To obtain ME coupling measurement, the sample has been poled by applying a high voltage across the thickness of the sample. The sample have been placed in a dc H and ac H , where the ac field is superimposed on dc field, the dc H gives a pseudopiezomagnetic effect, which generates an ac electric field ($dE = dV/t$) across the sample, where, t is the thickness of the sample. ϵ has been estimated from an induced voltage dV . The maximum value of ϵ_{31} and ϵ_{33} are found to ~ 3.7 and ~ 1.04 mV/cmOe at H of 0 Oe and 5 Oe, respectively at af of 73 Hz. ϵ_{31} vs. H and ϵ_{33} vs. H show gradual decrease with increase in H . The observed variation of ϵ with H for both the configurations may be due to the variation of piezomagnetic coefficient with H of piezomagnetic material of the sample[12]. The decreasing value of ϵ with increasing H may be due to the decrease of the growth of the domains with the increasing H leads the decrement of value for both the configurations[13].

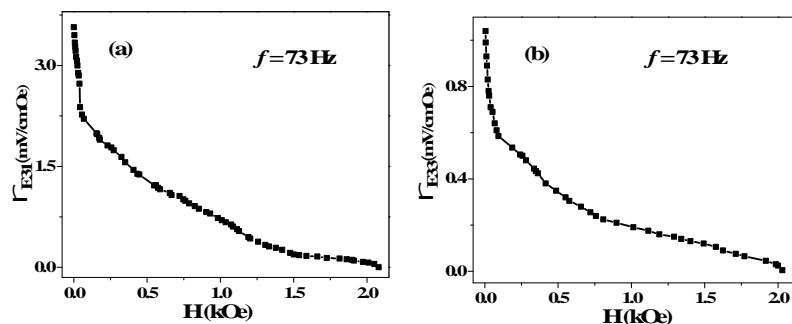


FIG 3: (a) ϵ_{31} vs. H , (b) ϵ_{33} vs. H for 0.2 NFO – 0.8 ZnO nanocomposites at $af = 73$ Hz.

CONCLUSIONS

In the present paper, we have investigated the magnetoelectric coupling and the ac electric transport properties by externally applied magnetic fields. Impedance spectroscopy study shows that the impedance is found to increase with increase in applied H . The calculated MI and MD are found to ~ 43 % at af of 800 Hz and ~ 34 % at af of 500 Hz. The maximum values of ϵ_{31} and ϵ_{33} are found to ~ 3.7 and ~ 1.04 mV/cmOe at H of 0 Oe and 5 Oe, respectively at af of 73 Hz.

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REFERENCES

- [1] G. Shan, X. Kong, X. Wang, Y. Liu, Surf. Sci. **582**(2005), 61-68.
- [2] A. Srivastava, Rashmi, K. Jain, Mater. Chem. Phys. **105**(2007), 385-390.
- [3] Z.L. Wang, J.Song, Science **80-312** (2006), 242-246.
- [4] S.k. Mandal, P. Dey, and T.K. Nath, Mater Sci. Eng., B **181** (2014), 70.
- [5] P. Dey, T.K. Nath, M.L.N. Goswami, and T.K. Kundu, Appl. Phys. Lett. **90**(2007), 162510.
- [6] S. K. Mandal, Rajesh Debnath, S. Singh, A. Nath, P. Dey and T. K. Nath, J. Magn. Magn. Mater. **443**(2017), 222.
- [7] P. Dey, R. Debnath, S. Singh, S.K. Mandal, J.N. Roy, J. Magn. Magn. Mater. **421** (2017), 132.
- [8] k.k.Mohaideen, P.A. Joy, Appl. Mater. Interfaces **4**, 6421(2012).
- [9] R. Debnath, P.Dey, S. Singh, J.N. Roy, S.K. Mandal and T.K. Nath, J. Appl. Phys. **118** (2015) 044104-8.
- [10] J.C. Maxwell, Electricity and Magnetism, vol. **1**, Oxford University Press, New York, p. 828 1993.
- [11] G. Catalan, Appl. Phys. Lett. **88**(2006), 102902.
- [12] G. Srinivisan, E.T. Rasmussen, A.A. Bush, K.E. Kametsev, V.F.Meshcheryakov, Y.K. Fetisov, Appl. Phys. A **78**, 721-728(2004).
- [13] S.Chakraborty, S.K. Mandal, R. Debnath, S. Singh, P. Dey and B. Saha, Materials today: Proceedings, **4**(2017), 5663-5666.